Synergistic Effect of Double-Shelled and Sandwiched TiO₂@Au@C Hollow Spheres with Enhanced Visible-Light-Driven Photocatalytic Activity

Jiabai Cai,†,§ Xueqing Wu,†,§ Shunxing Li,*,†,‡ Fengying Zheng,†,‡ Licong Zhu,† and Zhanghua Lai†

† College of C[he](#page-7-0)mistry and Enviro[nm](#page-7-0)ent, Minnan N[orm](#page-7-0)al University, Zhangzhou 363000, P. R. China

‡ Fujian Province Key Laboratory of Modern Analytical Science and Separation Technology, Minnan Normal University, Zhangzhou, 363000, P. R. China

S Supporting Information

[AB](#page-7-0)STRACT: [A novel appro](#page-7-0)ach for the fabrication of double-shelled, sandwiched, and nanostructured hollow spheres was proposed, using hydrotherm reaction and calcination. The negatively charged nanoparticles (e.g., Au, Ag, and Pt) could be adsorbed successively onto the positively charged hollow spheres (e.g., TiO₂, ZnO, and ZrO₂). The resulted nanocomposites $(TiO₂(@Au, as a proof-of-concept)$ were dispersed in glucose solution under hydrothermal conditions. After calcination, uniform double-shelled and sandwiched $TiO₂(\partial Au(\partial C$ hollow spheres were obtained and Au nanoparticles were sandwiched between the shell wall of $TiO₂$ and C. The samples were characterized by SEM, TEM, XRD, XPS, BET, and UV−vis DRS. The photocatalytic activity for the degradation of 4-nitroaniline was in the order of $TiO₂(\omega)$ Au@C > TiO₂@C > TiO₂/Au > P25. The visible-light photodegradation rate of 92.65% for 4-nitroaniline was achieved by $TiO₂(a)$

Au ω C, which exhibited an increase of 75% compared to Degussa P25 TiO₂. Furthermore, no deactivation occurred during catalytic reaction for three times, i.e., the TiO₂@Au@C microspheres exhibited superior photocatalytic stability. TiO₂@Au@C microspheres could also enhance the photocatalytic activity for hydrogen generation from methanol/water solutions. The synergistic effect of coupling $TiO₂$ hollow spheres with Au nanoparticles and C shell on photocatalytic performance was proved by us. The photoexcited electrons from Au nanoparticles could be captured by the conduction band of TiO₂ and then the electron–hole separation was improved. Moreover, both the visible light absorption and the affinity between TiO₂ and pollutants could be improved by the coexistence of carbonaceous materials, which could facilitate the photocatalytic interface reaction. KEYWORDS: double-shell sandwiched nanomaterials, sunlight, photodegradation, hydrogen production

1. INTRODUCTION

Over the past several years, sunlight-driven photocatalysis has been considered as an environment-friendly and efficient method for aromatic pollutants removal¹⁻³ and the production of clean and renewable hydrogen through water splitting.⁴ In the design of nanocomposite-based p[ho](#page-7-0)t[o](#page-7-0)catalysts, the major obstacles should be overcome, including electron−[ho](#page-7-0)le separation,⁵ light residence time in/on the photocatalyst, $6,7$ and the affinity between pollutants and photocatalyst.⁸ Such challenges [c](#page-7-0)an be unlocked by the innovation of the struct[ure](#page-7-0) and morphology of the photocatalyst.^{9−11} As an e[ff](#page-7-0)ective structure, hollow sphere has been used for solving the first two obstacles.12,13 Because of the unique [physi](#page-7-0)cal and chemical properties (e.g., large surface area, low density, and highly efficient [light](#page-7-0)-harvesting ability¹⁴), hollow spheres of metal oxides have attracted much attention because of their wide variety of applications, inclu[din](#page-7-0)g drug release systems, 15 heterogeneous catalysts, 16 waste removal, 17 the protection of light-sensitive molecules,¹⁸ and so on.

Recently, inorganic oxide supported Au catalysts have been used for the photodegradation of organic pollutants.^{19,20} Unfortunately, the supported Au catalysts with small sizes are generally unstable against thermal sintering and growing^{2[1](#page-7-0)} [so](#page-7-0) the catalytic activity of the supported Au catalysts decays rapidly. To avoid migration and aggregation of Au [nan](#page-7-0)oparticles (NPs), three kinds of strategies have been developed, including the replacement of monocomponent Au NPs with Au-based alloy NPs^{22} or hydroxyapatite composite,²³ the confinement of Au NPs within mesoporous supports, 24 and the encapsulation of [Au](#page-7-0) NPs into core−shelled microsp[her](#page-7-0)es.²⁵ Compared with the former two methods, the last one is [a m](#page-7-0)ore effective way, because the structure of core−shell has hi[gh](#page-7-0) thermal stability²⁶ and recyclability.²⁷ Among organic capping agents, carbonaceous materials have received particular

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attention, because of their advantages in visible-light absorption and organic pollutants adsorption, which can facilitate the photocatalytic reaction. Both $TiO₂(Q₂)$ carbon with dyad structure and hierarchical TiO₂@C hybrid hollow spheres exhibit an enhanced photocatalytic efficiency for visible-light photooxidation.^{28,29} Among the various inorganic oxides, including TiO₂,^{30–32} CeO₂,³³ ZnO,³⁴ and ZrO₂,³³ TiO₂ with a suitable band gap [of 3](#page-7-0).2 eV is proven to be the most widely used metal oxide[-ba](#page-7-0)[sed](#page-8-0) phot[oc](#page-8-0)atalys[t,](#page-8-0) because of [it](#page-8-0)s high photocatalytic activity, wide band gap, cost-effectivity, low toxicity, and excellent chemical stability. Despite good electronic conductivity of the carbon layer, the electron−hole recombination following excitation is still a challenge. Therefore, tailored nanostructures offer a new way for achieving this goal by facilitating electron−hole separation.

Herein, we design a general hydrothermal method to synthesize double-shelled and sandwiched $TiO_2(\partial A u/\partial C)$ hollow sphere, using $TiO₂$ hollow sphere as the template (Scheme 1 and Figure 1a). As a neutral surfactant, glucose has

Figure 1. (a) Schematic of the fabrication process for double-shelled and sandwiched $TiO₂(\partial Au/\partial C)$ hollow spheres. SEM images of (b) pristine $TiO₂$ hollow spheres (303 nm) and (c) double-shelled and sandwiched TiO₂@Au@C hollow spheres (323 nm).

been used for its nonpolar long carbon chain to control the growth of the nanocrystals. Our objective is to (a) offer a new way for facilitating electron−hole separation, using the Schottky barrier junctions of TiO₂ with Au (Scheme 2), (b) enhance the light-harvesting efficiency and mass transfer through the unique hierarchical mesoporous structure, (c) increase the accessible surface area of $TiO₂$, and (d) improve the affinity between photocatalyst and organic pollutants through carbon layer.

Scheme 2. Charge Transfer in Monodisperse Double-Shelled and Sandwiched TiO₂@Au@C Hollow Spheres

2. EXPERIMENTAL SECTION

2.1. Reagents and Chemicals. Tetrabutyl titanate (TBOT, J&K, 99%), tetrachloroauric (III) acid hydrate (AuCl₃·HCl·4H₂O, Sinopharm, 99%), trisodium citrate dihydrate $(C_6H_{12}O_7Na_3.2H_2O$, Xilong, 99%), D-glucose $(C_6H_{12}O_6·H_2O$, Xilong), and ethanol $(C_2H_5OH,$ Xilong, 99.7%) were purchased in China and used without further purification. Distilled water was employed as the polymerization medium.

2.2. Synthesis of Au Nanoparticle.³⁶ An aqueous solution of $HAuCl₄·3H₂O$ (0.1 wt %, 10 mL) was added to deionized water (100 mL) and heated to boiling point. A sodiu[m c](#page-8-0)itrate solution (1 wt %, 2 mL) was then added and the resulting mixture was kept for 15 min under stirring. Upon cooling to room temperature, the Au NPs were separated from the solution by centrifugation and redispersed in water (10 mL).

2.3. Synthesis of Monodisperse Polystyrene Spheres. The monodisperse polystyrene (PS) spheres were prepared by the interpolymerization of methyl acrylate and styrene, according to a
modified method.³⁷ The reactor used included a double-neck roundbottom flask, mechanical agitation, temperature controller, reflux condenser, and N_2 N_2 agitation. After replacing the air inside the reactor with N_2 , the reaction system was kept under N_2 until the polymerization was completed. After adding distilled water (85 mL), styrene (10.47 mL), and methyl acrylate (0.52 mL), the reactor was heated and kept at 70 °C. Na₂S₂O₈ solution (30 g/L, 5 mL) was added to start the polymerization process for 24 h, and then PS latex (ca. 10% solid content) was collected and used as templates.

2.4. Synthesis of $TiO₂$ Hollow Spheres. TBOT (0.18 mL) was dissolved in ethanol (10 mL), in which 0.25 g of PS dispersed in the mixture of water (0.15 mL) and ethanol (72 mL) was added. After refluxing at 80 °C for 4 h, the obtained $PS@TiO₂$ nanocomposite was washed three times with ethanol, dried in vacuum at 60 °C for 6 h, heated with a rate of 5 $\mathrm{C/min}$, and maintained at 550 C (i.e., calcination temperature) for 3 h to remove the PS core. Single-shelled TiO2 hollow spheres were finally obtained.

2.5. Synthesis of $TiO₂/Au$ Hollow Spheres. The prepared Au NPs in aqueous solution (3 mL) were mixed with water (30 mL) and this mixture was stirred for 10 min at room temperature. Subsequently, hollow TiO₂ (0.05 g) was added and the resulting mixture was kept for

10 h under stirring. The resultant solid product was centrifuged, washed three times with ethanol, and dried in vacuum at 60 °C for 6 h.

2.6. Synthesis of TiO₂@C Hollow Spheres. Glucose (1 g) was mixed with water (30 mL) and the reaction mixture was stirred for 30 min at room temperature. Subsequently, hollow $TiO₂$ (0.05g) was added and the resulting mixture was kept for 30 min under stirring. The mixture was transferred to Teflon-lined stainless steel autoclaves and heated to 180 °C for 3 h. TiO₂@C composite particles were obtained after centrifugation and washing procedures.

2.7. Synthesis of TiO_2 @Au@C Hollow Spheres. Hollow TiO₂ was mixed with water (30 mL) and the reaction mixture was stirred for 30 min at room temperature. Subsequently, the prepared Au NPs in aqueous solution (3 mL) were added and the resulting mixture was kept for 30 min under stirring. Glucose (1 g) was added and stirred for another 30 min at room temperature. The mixture was transferred to Teflon-lined stainless steel autoclaves and heated to 180 °C for 3 h. The obtained $TiO₂(\partial Au(\partial C)$ particles were washed by centrifugation/ redispersion cycles with ethanol for three times and dried in vacuum at 60 °C for 6 h. Finally, the spontaneously formed carbon shells were used as the stern layers to generate double-shelled and sandwiched $TiO₂(\partial Au/\partial C)$ hollow spheres through calcination in argon atmosphere.

2.8. Characterization. After being coated with a thin Pt layer, the samples were tested by a Gemini microscope (Hitachi, S-4800, Japan) at an accelerating voltage of 10 kV to obtain their scanning electron microscopy (SEM) images. Transmission electron microscopy (TEM) was conducted on a Tecnai G2 F20 U-TWIN (FEI, USA) under the acceleration voltage of 200 kV. Samples for TEM were prepared by drying a drop of dilute nanocomposite suspension on a carbon grid. Samples were also analyzed by energy dispersive X-ray spectroscopy (EDX) on the same TEM to disclose the independent existence of C shell and TiO₂ shell, i.e., double-shell in TiO₂@Au@C hollow spheres. Structural properties of the nanocomposite were investigated by X-ray diffraction (XRD, D/MAX-TTRIII(CBO), Rigaku Corporation, Japan) patterns with monochromatic Cu K α radiation (40 kV, 30 mA) in the range of 20°−80°. After being mixed with KBr, the samples were measured by a Thermo Nicolet FT-IR iS10 spectrometer (USA) and then Fourier-transform infrared (FTIR) spectra were obtained. The nanocomposite was dried in a vacuum for 12 h and then its thermogravimetric analysis (TG) was carried out a TG 2009F1 analyzer (NETZSCH, Germany) in the range of 25−900 °C at a heating rate of 15 K/min under nitrogen atmosphere. Fine BaSO₄ powder was used as a standard for the baseline, and the UV−vis absorption spectra of TiO₂, TiO₂/Au, TiO₂@C, and TiO₂@Au@C hollow spheres were recorded on a UV−vis 2550 spectrophotometer (Shimadzu). X-ray photoelectron spectroscopy (XPS, Thermal Scientific K-Alpha XPS spectrometer) was used to investigate the distribution of Ti, Au, and C in TiO₂@Au@C hollow spheres. Using an ASAP2020 instrument, BET surface area was determined via N_2 adsorption at 77 K over a relative pressure (P/P_0) range from 0.01 to 0.99.

2.9. Photocatalytic Activity Measurements. The photocatalysts (30 mg of P25, TiO₂, TiO₂/Au, TiO₂@C, or TiO₂@Au@C) with aromatic pollutant (4-nitroaniline or 4-nitrophenol, 5×10^{-3} g/L) were stirred in the dark for 1 h, and the adsorption/desorption equilibrium between pollutants and hollow spheres was reached. The photocatalytic performance was then tested under visible light irradiation at room temperature. A Xenon Short Arc Lamp (350 W) was used as light source, and the irradiation spectrum was cut off below 420 nm using a Pyrex glass for the simulation of sunlight.³⁸ After photodegradation for different time intervals, the samples were taken out of the aqueous dispersions, the photocatalysts were remov[ed](#page-8-0) by filtration, and the pollutant concentration was determined by spectrophotometry. The degradation rate was calculated using eq 1.

$$
Degradation rate = 100\% \times (C_0 - C)/C_0 \tag{1}
$$

Here, C_0 and C are the concentrations of the pollutant in the solution before and after visible light irradiation, respectively.

The first-order kinetic equation (eq 2) was used to fit the experimental data:

$$
\ln(C_0/C) = k_{\rm app} \times t \tag{2}
$$

where k_{app} is the reaction rate constant and t is the reaction time.³⁹

3. RESULTS AND DISCUSSION

3.1. Morphology and Structure of the Double-Shelled and Sandwiched TiO₂@Au@C Hollow Spheres. Aligned, double-shelled, and sandwiched $TiO₂(\partial Au(\partial C \text{ hollow spheres})$ were prepared on the template of $TiO₂$ hollow spheres by a one-step hydrothermal process (Experimental Section). In our work, the well-defined hollow $TiO₂$ submicrospheres could be easily obtained via a seed-assiste[d high-temperature ca](#page-1-0)lcination method, according to our previous work (Figure 1b).⁴⁰ In the pH range of 2–5, the outer layers of $TiO₂$ hollow spheres had positive surface charges (seen in Supporting Inf[orm](#page-1-0)a[tio](#page-8-0)n (SI) Figure S1),⁴⁰ which could enhance the affinity between $TiO₂$ and gold sol (negatively charged) 41 [by electrostatic adsorp](#page-7-0)tion, and then t[he](#page-8-0) surface coverage and adsorption rate of Au NPs onto the $TiO₂$ hollow spheres [w](#page-8-0)ere increased. Meanwhile, $TiO₂(\omega)$ Au hollow spheres could be prepared by this electrostatic adsorption process. Under high temperature and high pressure of hydrothermal treatment, the condensation of glucose in solution led to the formation of amorphous carbon shell, in which the Au NPs were located in the sandwich layer. After calcination in argon atmosphere, the spontaneously formed carbon shells were used as the stern layers to generate double-shelled and sandwiched $TiO_2@Au@C$ hollow spheres. The surface morphology and size distribution of $TiO₂$ and $TiO₂(\partial Au(\partial)C)$ hollow spheres are shown in Figure 1b and 1c, respectively. After coating Au NPs and C shell onto $TiO₂$ shell, the diameter of hollow spheres was increased from [30](#page-1-0)3 to [32](#page-1-0)3 nm. A smooth carbon shell was formed on the $TiO₂$ surface by a glucose-assisted hydrothermal method (for details, see SI Figures S2 and S3). The photocatalytic activity for the degradation of 4-nitroaniline was in the order of $TiO₂(\partial Au/\partial \theta)$ $C > TiO₂/\omega > TiO₂/Au > P25$. The visible-light photodegradation rate of 92.65% by TiO₂@Au@C was achieved, which exhibited a 75% increase compared to Degussa P25 $TiO₂$, because a Schottky barrier was assumed to form at the TiO₂−Au interface and the TiO₂ conduction band as electron sink could reduce the recombination of photoinduced electrons and holes. This result was similar to that of Au@TiO₂ core− shell hollow spheres.⁴²

In agreement with the SEM data (Figure 1c), the TEM images (Figure 2a) [rev](#page-8-0)ealed that the produced $TiO₂(\partial Au(\partial)C)$ were hollow and porous. A close inspection of t[he](#page-1-0) TEM image of TiO₂@Au@[C](#page-3-0) hollow spheres (Figure 2c) revealed the presence of individual nanoparticles on a thick and layered shell. [T](#page-3-0)he morphology and structure of TiO_2 , TiO_2/Au , $TiO_2@$ C, and $TiO₂(\partial Au(\partial)C)$ hollow spheres was characterized by SEM and TEM (SI Figure S4). The HRTEM image of the nanocomposite showed three types of intimately contacted lattice fringe[s,](#page-7-0) confirming the formation of the junctions between $TiO₂$, Au, and C (Figure 2b). The lattice spacing of 0.351 nm was the interplanar distance between adjacent (101) crystallographic planes of $TiO₂$, whi[le](#page-3-0) the lattice spacing of 0.24 nm matched that of the (111) plane of Au NPs. Furthermore, the coated carbon layer with a thickness of ca. 10 nm (Figure 2b) created a smooth, highly conductive layer over the entire nanospheres surface. The polycrystalline nature of these [in](#page-3-0)dividual nanoparticles was also confirmed by the SAED measurements (Figure 2d). Several important features could be recognized. The prepared microspheres had rather uniform

Figure 2. TEM (a and c), HRTEM (b), and SAED (d) images of double-shelled and sandwiched $TiO_2@Au@C$ hollow spheres.

diameters of 323 nm (Figure1c). By changing the volume of added Au NPs (3, 1.5, 1, or 0.5 mL), the content of Au NPs on the microspheres could be tu[ne](#page-1-0)d (SI Figure S5). The diameter of Au NPs in the sandwich layer was about 16−20 nm (SI Figure S6). The HRTEM image f[urth](#page-7-0)er indicated that each Au NPs was highly crystalline with the interplanar spacing of 0.[24](#page-7-0) nm, corresponding to the (111) planes of Au NPs. It was noted that the lattice spacing of Au NPs could not be changed clearly due to hydrothermal and calcination process (Figure 2c). In our system, the well-defined hollow microspheres were easily obtained in the presence of gold sol. A plausible explanation was that gold sol could accelerate the formation process of $TiO₂(\partial Au(\partial)C)$ solid microspheres. Meanwhile, the thickness of C shell could be easily adjusted by changing the quantity of glucose $(1, 2, 3, \text{ or } 4 \text{ g})$ in the reaction mixture (SI Figures S7 and S8).

The hollow spheres of [TiO](#page-7-0)₂, TiO₂/Au and TiO₂@Au@C were investigated by XRD analysis and the results are shown in Figure 3. Six peaks of TiO₂ were found at $2\theta = 25.22$, 37.78, 47.94, 54.15, 54.96, and 62.69, which were indexed to the (101), (004), (200), (211), (105), and (204) planes of TiO_2 , respectively, and these results were in good agreement with the standard pattern of anatase (JCPDS 21-1272). The diffraction peaks of $TiO₂/Au$ microspheres had a consistent position with pure $TiO₂$ submicrospheres. The constant peak position indicated the phase structure of $TiO₂$ had not been changed during the Au coating process. Furthermore, the Au coating process did not provide any structural or morphological transformations as demonstrated by SEM and TEM results (Figures 1 and 2, respectively). Interestingly, after the second layer-by-layer deposition (i.e., carbon coating), the XRD pattern o[f](#page-1-0) TiO₂@Au@C still only demonstrated the characteristic of anatase $TiO₂$ phase, because partial carbon atoms from $TiO₂(\partial Au/\partial C)$ existed in an amorphous phase and the content of carbon was so limited that strong diffraction peaks could not be observed.

Figure 3. XRD pattern of $TiO₂$, $TiO₂/Au$, and double-shelled and sandwiched $TiO₂(\partial Au(\partial C \text{ hollow spheres}).$

The full range XPS spectrum of $TiO₂(@Au@C$ hollow spheres is shown in Figure 4a. The XPS peaks of Ti $2p_{3/2}$ and $2p_{1/2}$ were located at about 458.5 and 464.2 eV with a good symmetry (Figure 4b), ind[ica](#page-4-0)ting that the chemical valence of Ti was $+4$ in octahedral coordination with oxygen.²⁹ Moreover, the chemical envir[on](#page-4-0)ments for Ti and O_L were not changed, strongly suggesting that carbon and nitrogen ori[gin](#page-7-0)ated from TBOT did not enter the $TiO₂$ crystal lattice under the mild solvothermal conditions. The XPS spectra of TiO₂@Au@C (Figure 4c) show the peaks of Au $4f_{7/2}$ and Au $4f_{5/2}$ centered at 84.2 and 87.8 eV, respectively. The spin energy separation of 3.6 eV [wa](#page-4-0)s in good agreement with reported data of Au $4f_{7/2}$ and Au $4f_{5/2}$ in Au NPs.³⁶ Figure 4d shows the C 1s peak from XPS, which corresponded to the carbon layers in the TiO₂@ Au@C hollow spheres. [No](#page-8-0) obvio[us](#page-4-0) peaks were detected around 281.5 and 283 eV, indicating that the carbon originated from glucose was not doped into the $TiO₂$ lattice.

The diffuse reflectance UV−vis spectra of the prepared photocatalysts are shown in Figure 5. The UV absorption band at ~400 nm, corresponding to the band gap of 3.2 eV,⁴³ could be attributed to the band-edge ab[so](#page-4-0)rption of anatase $TiO₂$.⁴⁴ After coating C shell onto hollow $TiO₂$ $TiO₂$ and hollow $TiO₂/Au$, their absorbances in the region of 300−400 nm were affect[ed](#page-8-0) by the band absorption of the amorphous carbon (shown in Figure 5). The absorption features from 360 to 600 nm arose from the band absorption of the amorphous carbon.⁴⁵ The absorp[tio](#page-4-0)n of Au NPs and amorphous carbon was in the range of 500−600 nm³⁶ and 360−600 nm, respectively. [T](#page-8-0)heir absorption spectra were seriously overlapping. Furthermore, in the TiO₂@Au[@C](#page-8-0) hollow spheres, the content of C was higher than that of Au, and the absorption of C was stronger than that of Au. No obvious difference in the UV−vis spectra between TiO₂@C and TiO₂@Au@C was observed. The light absorption of $TiO₂(\partial Au(\partial)C)$ hollow spheres was stronger than that of $TiO₂$ hollow sphere in sunlight light region (Figure 5), i.e., the visible-light harvesting efficiency of $TiO₂(\partial Au(\partial C))$ hollow spheres was improved.

Besides high photostability and excellent water-solubility, l[ow](#page-4-0) cytotoxicity was important for $TiO₂(@Au@C$ hollow spheres as safe nanomaterials. The cytotoxicity of $TiO₂(\partial Au(\partial C \text{ hollow}$ spheres was evaluated using a standard cell viability assay (i.e., 3-[4,5-dimethylthiazol-2-yl]-2,5-diphenyltetrazolium bromide

Figure 4. Full range of the XPS spectrum (a), XPS peaks of Ti 2p (b), Au 4f (c), and C 1s (d) of double-shelled and sandwiched TiO₂@Au@C hollow spheres.

Figure 5. Diffuse reflectance UV-vis spectra of P25, TiO₂, TiO₂/Au, TiO₂@C, and double-shelled and sandwiched TiO₂@Au@C hollow spheres (the 500−600 nm is gold plasmonic band).

(MTT) assay). HeLa cells exposed to TiO₂@Au@C with different concentrations $(0-200 \ \mu g \ mL^{-1})$ for 24 h and the influence of TiO₂@Au@C concentrations on cell viability was tested. As shown in Figure 6, 79.12% of HeLa cells retained viability after being exposed to 200 μ g mL⁻¹ of TiO₂@Au@C. The form of HeLa cell was not so much obviously changed (SI Figure S10). Even at the highest concentration tested (200 μ g mL⁻¹), the cytotoxicity of TiO₂@Au@C was not significant [in](#page-7-0) our MTT assay, which implied that the double-shelled and sandwiched TiO₂@Au@C hollow spheres were lowly toxic and environmentally friendly.

3.2. Evaluation of Photocatalytic Activity. To investigate the influence of the photocatalyst structure on catalytic

Figure 6. HeLa cell viability after incubating with double-shelled and sandwiched TiO₂@Au@C hollow spheres for 24 h and quantitative assays by standard MTT method.

activity, five kinds of photocatalysts, including P25, hollow TiO₂, hollow TiO₂(∂ C, hollow TiO₂/Au, and double-shelled and sandwiched TiO₂@Au@C hollow spheres, were used for the photodegradation of 4-nitroaniline and 4-nitrophenol under simulated sunlight irradiation for 180 min. The results are shown in Figure 7. Only 17.61% of 4-nitroaniline could be photodegraded by P25. The photocatalytic activity of $TiO₂(@)$ Au@C was higher [th](#page-5-0)an that of P25, TiO₂, TiO₂@C, and TiO₂/ Au. The photodegradation rate of 4-nitroaniline by $TiO₂(\partial Au/\partial \theta)$ C was 92.65%. Meanwhile, the linear relationship between $ln(C_0/C)$ and t indicated that the photodegradation reactions followed pseudo-first-order kinetics (Figure 7) with rate constants (0.0012, 0.0004, 0.0050, 0.0046, and 0.015 min[−]¹) and correlation coefficients R^2 (0.987, 0.987, 0.[99](#page-5-0)9, 0.995, and

Figure 7. Linear transform $\ln(C_0/C) = f(t)$ of the kinetic curves of (a) 4-nitroaniline or (b) 4-nitrophenol photodegradation by P25, hollow TiO₂, hollow TiO₂@C, hollow TiO₂/Au, and double-shelled and sandwiched TiO₂@Au@C hollow spheres under visible light irradiation.

0.999) for P25, TiO₂, TiO₂@C, TiO₂/Au, and TiO₂@Au@C, respectively. For comparison, the same procedure was also performed for 4-nitrophenol (Figure 7, Table 1, and SI Table

Table 1. Visible Light Photodegradation Rate Cons[tan](#page-7-0)t k (min^{-1}) for P25, TiO₂, TiO₂@C, TiO₂/Au, and Double-Shelled and Sandwiched TiO₂@Au@C Hollow Spheres

S1). The relative photocatalytic activity of photocatalysts for aromatic pollutant degradation was in the order of $TiO₂(\partial Au/\partial \theta)$ $C > TiO₂/\omega > TiO₂/Au > P25$. The photocatalytic performance of hollow $TiO₂$ was improved by the coupling of Au and C (Figure 7 and Table 1). This synergistic effect could be attributed to the following reasons. After coating carbon shell onto $TiO₂$ hollow spheres, the affinity between photocatalyst and aromatic pollutants could be improved. Compared with $TiO₂$ hollow spheres, the adsorption ratio of $TiO₂(\partial Au/\partial C)$ for 4-nitroaniline and 4-nitrophenol in 60 min was increased 1.13% and 5.80%, respectively (Figure 8). Certainly, the surface coverage of aromatic pollutants on the $TiO₂(\partial Au/\partial C)$ was more than that on $TiO₂$ hollow spheres. Energy could be transferred from aromatic pollutants to the $TiO₂$ conduction band. The back electron could also be transferred from Au NPs to the $TiO₂$ conduction band. The photodegradation rate of 4-nitroaniline was used as a measure of the efficiency of these prepared nanocatalysts in producing •OH radicals. The visible-light photodegradation rate of TiO₂@Au@C was 92.65%, which exhibited a 30% increase compared with TiO₂@C. The electron–hole separation could be facilitated through the unique hierarchically mesoporous structure. On the other hand, a Schottky barrier was assumed to form at the TiO₂−Au interface, and the TiO₂ conduction band as electron sink could reduce the recombination of photoinduced electrons and holes. The N_2 gas has been used by us to remove the oxygen in the water and the photodegradation rate was reduced (SI Figure S11). The result indicated that the photocatalytic activity of TiO₂@Au@C could be suppressed by the decrease o[f o](#page-7-0)xygen concentration. This result proved that the •OH radicals, which were mainly depending on the electron-transfer mediation, played an important role in the photocatalysis process. The specific surface area, pore volume, and average pore size of $TiO₂(\partial Au(\partial C \text{ hollow spheres were}$

Figure 8. Adsorption rate of (a) 4-nitroaniline and (b) 4-nitrophenol on P25, hollow TiO₂, hollow TiO₂/ ω C, hollow TiO₂/Au, and double-shelled and sandwiched $TiO₂(\partial Au(\partial)C)$ hollow spheres under dark.

62.95 $\mathrm{m^2~g^{-1}}$, 0.1452 $\mathrm{cm^3~g^{-1}}$, and 1.352 nm, respectively (SI Figure S12 and Table S2), indicating TiO_2 @Au@C hollow spheres were also typical mesoporous materials. The stability [of](#page-7-0) TiO₂@Au@C was examined by the degradation of 4-nitroaniline during a three-cycle experiment. After three cycles, the photocatalytic activity decreased slightly, which indicated that $TiO₂(\partial Au/\partial C)$ exhibited good photocatalytic stability (SI Figure S13). After three cycles experiment of photodegradation, the hollow structure of $TiO₂(\partial Au/\partial C)$ still remaine[d,](#page-7-0) which indicated that the structure stability and mechanical strength of TiO₂@Au@C were good (SI Figure S14).

The time vs hydrogen production for as-prepared samples is shown in Figure 9. Under irra[dia](#page-7-0)tion with the light beam ($\lambda \geq$

Figure 9. Visible-light photocatalytic hydrogen generation by hollow TiO₂, hollow TiO₂/Au, hollow TiO₂@C, and double-shelled and sandwiched TiO₂@1.43%Au@C hollow spheres in methanol–water system.

420 nm), the values of hydrogen production rate constant (r_{H2} , obtained as the slope of the linear part of the curves in Figure 9 and SI Figure S15) are collected in Table 2 and Table 3. Under

Tab[le](#page-7-0) 2. Hydrogen Production Rate (r_{H2}) in the Visible-Light Photocatalytic Splitting of Methanol−Water System

sample	$\int_{\mu}^{r_{\text{H2}}} \mu$ = $\int_{\mu}^{r_{\text{H2}}}$ =	relative rate
P ₂₅	0	
hollow TiO ₂	Ω	
hollow $TiO2(\omega)C$	Ω	
hollow $TiO2/Au$	193.2	1
double-shelled and sandwiched $TiO2(a)1.43%$ Au@C	347.1	1.8

Table 3. Hydrogen Production Rate (r_{H2}) in the Visible-Light Photocatalytic Splitting of Methanol−Water System

visible light irradiation, the hydrogen production rate of doubleshelled and sandwiched TiO₂@1.43%Au@C hollow spheres was 319.5 μ mol h⁻¹ g⁻¹ at first hour, which was higher than that of P25 (Table 2). The hydrogen production rate was found to be fairly linear with light irradiation time from 1.0 to 5.0 h. We also tested the P25, hollow TiO₂, and hollow TiO₂@C for photocatalytic hydrogen production at the same conditions, and no hydrogen could be determined. The improvement on the photocatalytic hydrogen production activity might be due to the fast separation of the photoinduced electron/hole pairs for the stability of Au NPs.^{4,46,47}

The reaction pathway (R1) was as follows:

$$
CH_3OH \to H_2 + CH_2O \tag{R1}
$$

Reaction 1 was a multistep reaction:

$$
Au + hv \to e^- + h^+ \tag{R1a}
$$

$$
CH_3OH + 2h^+ \stackrel{Au}{\rightarrow} CH_2O + 2H^+ \tag{R1b}
$$

$$
2H^{+} + 2e^{-} \rightarrow H_{2} \tag{R1c}
$$

The role of Au was important, because photoexcited electrons from Au NPs could be trapped by the $TiO₂$ conduction band and some of them were subsequently transferred to hydrogen ions (Scheme 3).^{48,49} By changing

Scheme 3. Mechanism of Methanol Photo[cleav](#page-8-0)age over Photogenerated Electrons (e¯) and Holes (h $^{\mathrm{+}}$) Migration to $TiO₂$ and Au of the Monodisperse Double-Shelled and Sandwiched $TiO₂(\partial Au(\partial C)$ Hollow Spheres

the volume of added Au NPs (3, 1.5, 1, or 0.5 mL), the content of Au NPs on the microspheres could be tuned (SI Figure S5). The hydrogen production rates of $TiO₂(\partial Au(\partial)C$ with different contents of Au NPs are listed in SI Figure S15. [Th](#page-7-0)e hydrogen production rates of $TiO₂(\partial Au/\partial C)$ were different with different cont[en](#page-7-0)ts of Au NPs. The hydrogen production rate of $TiO₂(@)$ 1.43%Au@C was the highest, i.e, 1.43% was the optimal content of Au NPs in TiO₂@Au@C. A further increase in Au loading (i.e., more than 1.43%) could lead to the increase of Au particle size (SI Figure S5) and resulted in the decrease of photocatalytic hydrogen production rate (Table 3). This decrease coul[d b](#page-7-0)e attributed to a reduction in the available metal−support sites where both metal and support were needed for the reaction to occur.⁴⁸

4. CONCLUSIONS

In summary, we proposed a simple approach for the fabrication of TiO₂@Au@C hollow spheres and other heterojunction nanocomposites (e.g., TiO₂@Ag@C, TiO₂@Pt@C, ZnO@ Au@C, ZnO@Ag@C, ZnO@Pt@C, etc.) with uniform

double-shelled and sandwiched structure, which could be used for environmental cleanup.

■ ASSOCIATED CONTENT

S Supporting Information

SEM and TEM images, XRD pattern, EDX data, TG data, FTIR spectra, nitrogen adsorption−desorption isotherm, surface area, and porosity measurements of $TiO₂(\partial Au(\partial)C$ hollow spheres, correlation coefficient R^2 . This material is available free of charge via the Internet at http://pubs.acs.org/.

E AUTHOR INFORMATI[ON](http://pubs.acs.org/)

Corresponding Author

*Tel.: +86 596 2591395. Fax: +86 596 2591395. E-mail: lishunxing@mnnu.edu.cn.

Author Contributions

 $\mathrm{^{\$}}$ [Xueqing Wu and Jiabai](mailto:lishunxing@mnnu.edu.cn) Cai contributed equally to this work.

Notes

The authors declare no competing financial interest.

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